

High Q factor Electrically Injected Green Micro Cavity

Yang Mei , Yan-Hui Chen , Lei-Ying Ying, Zhi-Wei Zheng , Hao Long , and Bao-Ping Zhang

Abstract—Green micro-cavity (MC) with a quality factor (Q) exceeding 6000 was demonstrated under current injection. The MC consists of two dielectric DBRs and InGaN/GaN quantum wells in between. To form a lateral optical confinement (LOC) structure, a low-index thin AlN layer was used to replace part of the p-GaN. The device showed multi-longitudinal mode emission, from 460 to 540 nm, and higher-order lateral confined modes were clearly observed. By optimizing the cavity fabrication processes and lateral optical confinement, the linewidth of resonant mode is as narrow as 0.082 nm, indicating a high Q value of 6039 in green spectral region. This is the highest value in electrically injected GaN-based MC to the best of our knowledge. In addition, 3D confined optical states were observed and studied via angle resolved measurement for the first time in an electrically injected GaN-based MC. The emission characteristics of devices as a function of cavity length, as well as the loss mechanism inside the cavity were also systematically analyzed.

Index Terms—3D confined optical states, electrically injection, high Q factor, loss analyses, micro cavity.

I. INTRODUCTION

HIGH Q factor MCs can strongly confine photons in small volumes and a strong light-matter interaction can be obtained, thus attracting much attention in various fields, including photonics [1]–[3], telecommunications [4], quantum information [5], and cavity quantum electrodynamics. GaN-based semiconductors, including those of GaN, AlN, InN and their mixed alloys, are featured with wide bandgap and tunable emission spectrum covering from ultraviolet to near infrared [6], [7]. Besides, GaN-based materials have high oscillator strength (usually 10 times higher than GaAs) and large exciton binding energies [8]. These properties suggest that GaN-based MC with a high Q factor is an ideal platform for not only high efficient practical optical-electronic devices [9], [10], but also fundamental researches concerning light-matter interaction and cavity quantum electrodynamic (CQED) effect up to room temperature [11]–[14]. For the applications mentioned above, optical resonators

with high quality, or equivalently, a high Q factor, are required. The cavity Q factor, which is defined as the totally stored energy divided by the energy loss over one radian of the oscillating cycle (2π radian), is an important parameter to evaluate the quality of resonators [15], [16]. The high Q resonances mean that a lot of energy can be stored in the waveguide layer. Exploring the methods to increase the cavity Q factor has been one of the main research directions of semiconductor MC. The Q factor of silicon and GaAs based MCs at telecom range has reached a very high value, exceeding 1000000 [14]. For GaN-based MCs designed for shorter wavelengths, namely from the visible to the ultraviolet spectral range, the Q factor is much smaller due to the increased scattering loss and the inferior crystal quality. The Q factors of the state-of-the-art optically pumped GaN-based MCs are in the order of 10000, including the Q factor of ~ 18000 in Fabry-Perot (FP) planar cavity [17], [18], ~ 14000 in defective photonic crystal cavity [19], [20], and ~ 10200 in micro-disk cavity based on whispering-gallery modes (WGMs) [21], [22]. However, it is much more difficult for their electrically injected counterparts to realize such a high Q factor. At present, the highest cavity Q factor of electrically injected GaN-based MCs are ~ 3570 for planar cavity [23], ~ 800 for photonic crystal cavity [24], and less than 1000 for WGM cavity [25]–[27]. Electrically injected MCs have a more complicated structure to realize the current injection. The possible damages caused by the complicated fabrication processes, as well as the doping of the membrane might increase the internal loss and degrade the cavity Q factor [28].

In this work, we report an electrically injected green GaN-based planar MC with a high Q factor of 6039 by optimizing the fabrication processes and introducing a LOC structure. Two dielectric DBRs with high reflectivity were utilized and a sputtered AlN layer was used to confine both injected current and optical field. The device showed an emission spectrum covering from 460 to 540 nm, and the linewidth of the resonating mode is as narrow as 0.082 nm, indicating a high Q factor of 6039 in the green spectral region. This is the highest value in electrically pumped GaN-based MCs to the best of our knowledge. The loss mechanism in the cavity and the emission characteristics of devices with different cavity lengths were systematically analyzed. 3D confined optical states inside the cavity were also observed and studied through angle-resolved measurement. Due to the 3D optical confinement structure, the dispersion of optical states in momentum space is much more different from the typical continuous parabolic curve in a 2D planar MC. The mode dispersion is split into a series of discrete modes at different

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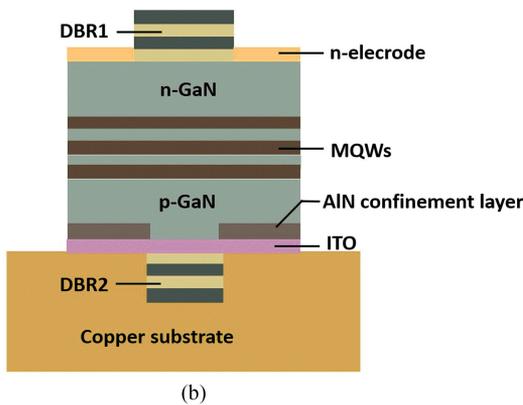
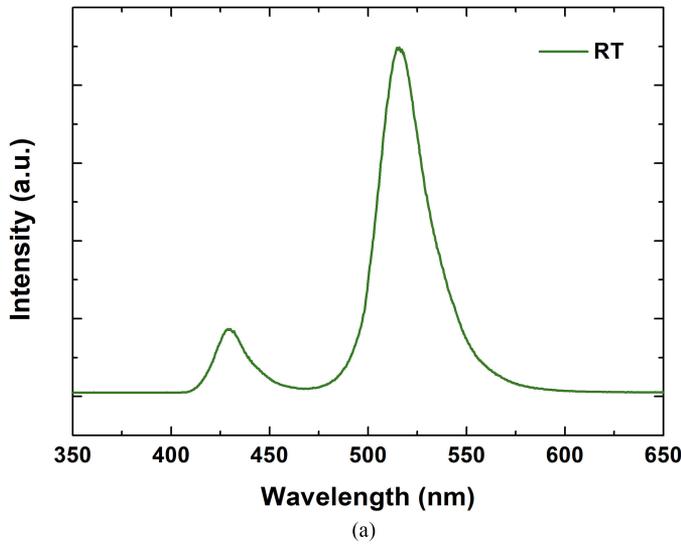


Fig. 1. (a) RT PL spectrum of the epitaxial wafer. (b) Cross-sectional schematic of the fabricated MC light emitter.

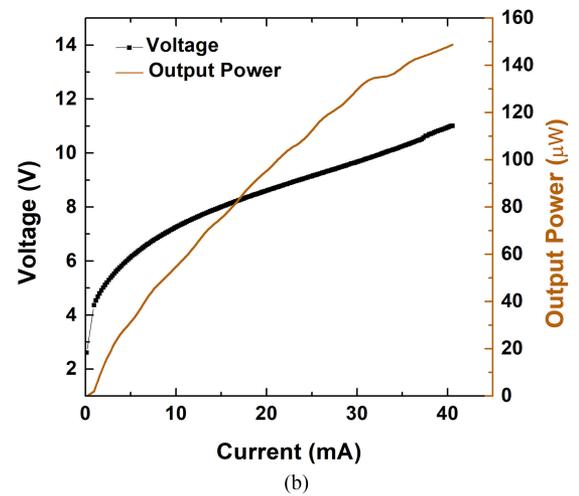
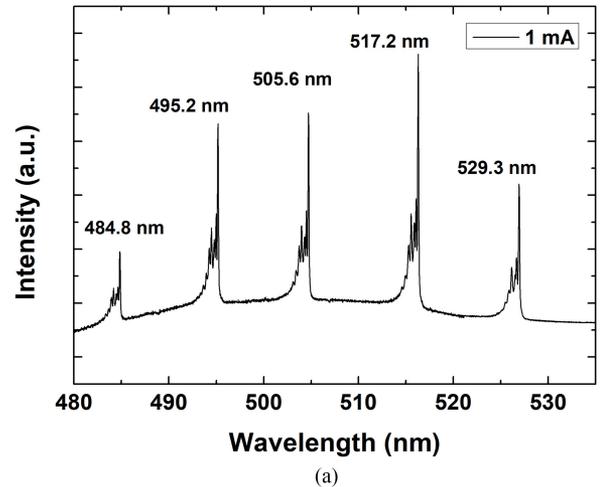


Fig. 2. (a) RT EL spectra of the device measured under 1 mA. (b) L-I-V characteristics of the device.

85 energy. The optical state density at these distinct energies can be
 86 much higher than that of a continuous dispersion. This “photonic
 87 dot” effect in this work is promising for the study of CQED at
 88 room temperature including the Purcell effect and Bose-Einstein
 89 condensation of cavity polaritons.

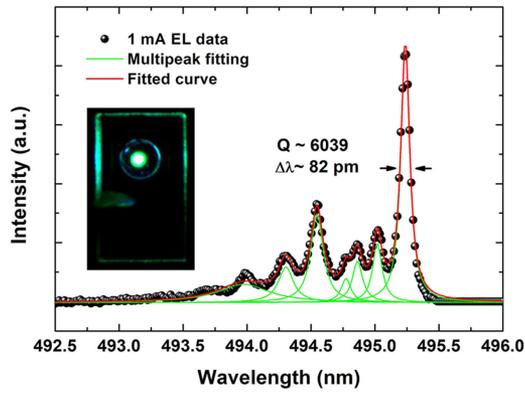
90 II. MATERIALS AND METHODS

91 The epitaxial wafer was grown on a (0001) c-plane patterned
 92 sapphire substrate by metal-organic chemical vapor deposition.
 93 The active medium consists of 3 pairs InGa_N/Ga_N pre-well (with low
 94 Indium content, ~ 0.1) and 16 pairs In_{0.25}Ga_{0.75}N/Ga_N (3/12 nm)
 95 QWs. The room-temperature (RT) photoluminescence (PL) spectrum
 96 of the epitaxial wafer is shown in Fig. 1(a). The spontaneous
 97 emission is centered at around 520 nm, and the small fringe peak
 98 at the higher energy comes from the pre-wells with low Indium
 99 content. The schematic structure of the device is illustrated in
 100 Fig. 1(b). The copper substrate and dual dielectric DBRs structure
 101 were realized by substrate transferring and laser lift-off (LLO)
 102 techniques, which is similar to our previous reports [29]–[31].
 103 The top and bottom DBRs are 10 and 12.5 pair of TiO₂/SiO₂,
 104 respectively. A

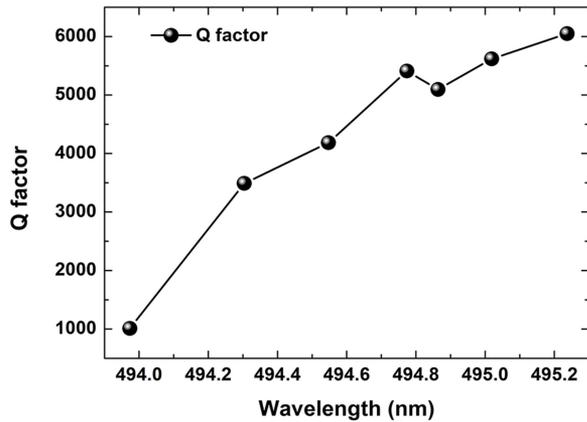
cylindrical mesa with a diameter of 7 μm was formed on p-GaN
 and a 60 nm thick sputtered AlN layer was inserted between
 p-GaN and ITO to laterally confine both optical field and
 injected current. Together with the dual DBRs, this lateral
 confinement layer can generate 3D confinement for the optical
 field, and effectively reduce the lateral optical loss. During
 device fabrication, the surface roughness of ITO and n-GaN
 were carefully controlled, and the root-mean-square roughness
 (RMS) of these cavity edge surfaces are 0.7 and 0.29 nm,
 respectively. This atomic-level flat cavity surface is essential
 to reduce scatter loss and guarantee a high cavity Q factor.

116 III. RESULTS AND DISCUSSION

117 Fig. 2(a) presents the RT electroluminescence (EL) spectra
 118 measured from the MC light emitter under 1 mA. The light
 119 emission of the device was collected by an objective lens
 120 (NA0.35, 10 \times) and then guided to the spectrometer. Multi-
 121 longitudinal mode emission, ranging from 460 to 540 nm,
 122 was observed from the spectra due to the relative long cavity
 123 length ($\sim 3.8 \mu\text{m}$) and the broad spontaneous emission
 spectrum of the QW wafer. The



(a)



(b)

Fig. 3. (a) High resolution spectrum of the 495.25 nm longitudinal mode. (b) Q factor of lateral modes with different mode order.

main longitudinal modes of the spectrum located at 484.8, 459.2, 505.6, 517.2, 529.3 nm, respectively. All the longitudinal modes show a multi-peak structure, which comes from the high-order lateral confined modes induced by the index-guiding effect of the buried AlN aperture (refractive index of AlN ~ 1.98 , GaN ~ 2.40 at 500 nm). The light-current-voltage (L-I-V) characteristic of the device is given in Fig. 2(b). The turn-on voltage is around 6.4 V and the maximum output power is 149 μ W under injected current of 40 mA.

The high-resolution spectrum of the 495.2 nm longitudinal mode is shown in Fig. 3(a). More than 7 higher-order lateral confined modes could be recognized and the fundamental mode is featured with the strongest intensity and a linewidth as narrow as 82 pm fitted by Lorentzian function. The Q factor calculated by $\lambda/\Delta\lambda$ is 6039, which is the highest value in electrically injected GaN-based MCs to the best of our knowledge. The high Q factor is attributed to the low loss in the cavity enabled by the high reflective DBRs, atomic level flat cavity end surfaces, and the lateral optical confinement. The effect of these items will be discussed later. The inset of Fig. 3(a) shows the near field emission pattern of the device under 1 mA. The bright emission spot locates pretty well in the center of the aperture, indicating that the injected current and the lateral optical field were effectively confined by the AlN aperture. Fig. 3(b) indicates the Q factors

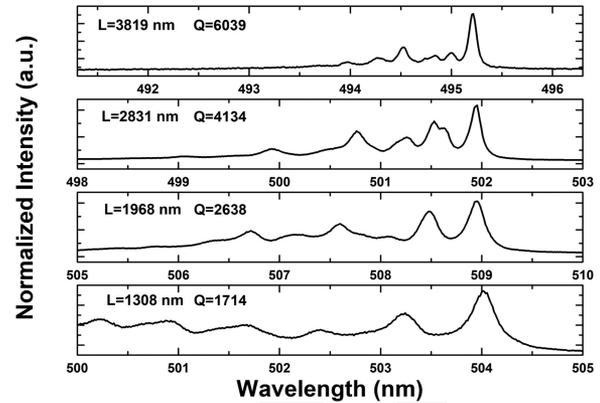


Fig. 4. Emission spectra measured from devices with different cavity length.

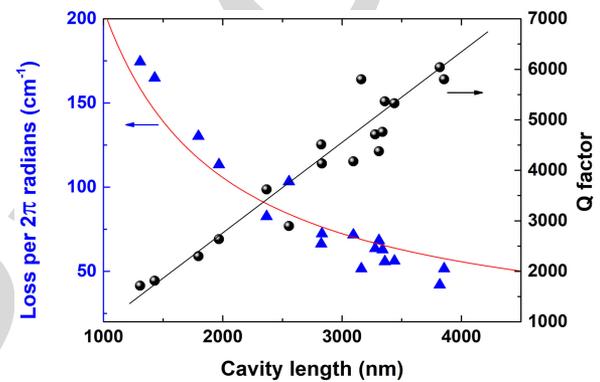


Fig. 5. Cavity Q factor and mode loss per 2π radians as a function of cavity length.

of different order 3D confined modes belongs to the 495.2 nm cavity mode as a function of their emission wavelengths (shorter wavelength corresponds to higher mode order). The Q factor is the highest for the fundamental mode and then decreases with the increase of mode order. This is consistent with the fact that the high-order lateral modes usually located near the edge of confinement aperture, resulting in a larger lateral optical leakage. Also, the diffraction angle of the higher-order modes is larger than the fundamental mode [32], thus more light could penetrate the DBRs and escape from the cavity.

Emission characteristics of devices with different cavity lengths were also investigated. Fig. 4 shows the emission spectra measured from devices with different cavity lengths of 1308, 1968, 2831, and 3819 nm, respectively. Emission from higher-order lateral modes can be clearly observed in all devices, but the linewidth of the fundamental mode decreases with the increase of cavity length. The Q factor of the fundamental emission mode increased from 1714 to 6039 when cavity length was increased from 1038 to 3819 nm. It is also worth to note that the intensity ratio between the fundamental mode and higher-order 3D confined modes is increased, and the latter is suppressed in a longer cavity. This is caused by the increased diffraction loss of higher-order modes which are featured with a larger diffraction angle in a longer cavity. Fig. 5 indicates the experimentally

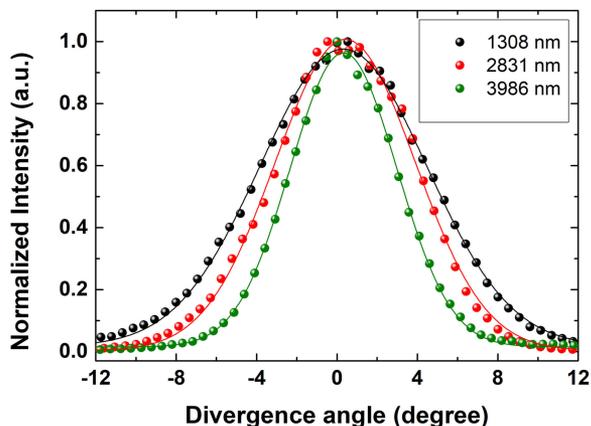


Fig. 6. Far-field image of the fundamental emission mode measured from devices with different cavity lengths.

172 measured Q factor of fundamental emission mode, as well as
 173 the calculated mode loss per 2π radians as a function of the
 174 cavity length. The cavity Q factor increases linearly with the
 175 cavity length. This is consistent with the definition of the Q
 176 factor, because the mirror loss and scatter loss take place only
 177 at the cavity edge surface, and the internal loss can be assumed
 178 approximately constant when the optical field propagates inside
 179 the cavity. In a longer cavity, the mirror loss and scatter loss are
 180 averaged to the total cavity when calculating the mode loss per
 181 2π radians, inducing the decrease of mode loss and increase of
 182 Q factor.

183 In addition to standard spectra measurement, far-field emis-
 184 sion characteristics of devices with different cavity lengths are
 185 also analyzed by using angle-resolved emission measurement.
 186 Fig. 6 shows the far-field image of the fundamental emission
 187 mode measured from devices with different cavity lengths of
 188 1308, 2831, and 3986 nm, respectively. The full-width at half-
 189 maximum far-field angle of the fundamental modes in these
 190 three devices are 9.8° , 8.2° , and 6.3° , respectively. Devices with
 191 shorter cavity length exhibit a larger divergence angle. Similar
 192 phenomenon has been observed in GaAs based VCSELs [33].

193 The distribution of 3D confined optical states in the momen-
 194 tum space was also studied through angle-resolved measure-
 195 ment, as shown in Fig. 7. Different from the typical parabolic
 196 type continuous mode dispersion in 2D planar MC without
 197 lateral confinement (white dashed curve), 3D confined optical
 198 states were clearly observed in this work. The dispersion of
 199 optical states in k space splits into a series of discrete modes
 200 at different energies. The state density of each mode as a function
 201 of $k_{in-plane}$ follows well the simple harmonic oscillator model
 202 in a parabolic potential well. This is the first observation and
 203 k space measurement of the 3D confined optical states in an
 204 electrically injected GaN-based MC. The optical state density at
 205 these distinct energies is much higher than that of a continuous
 206 dispersion, while there is no state distribution at other energy.
 207 This 3D optical confinement in MC with a high Q factor,

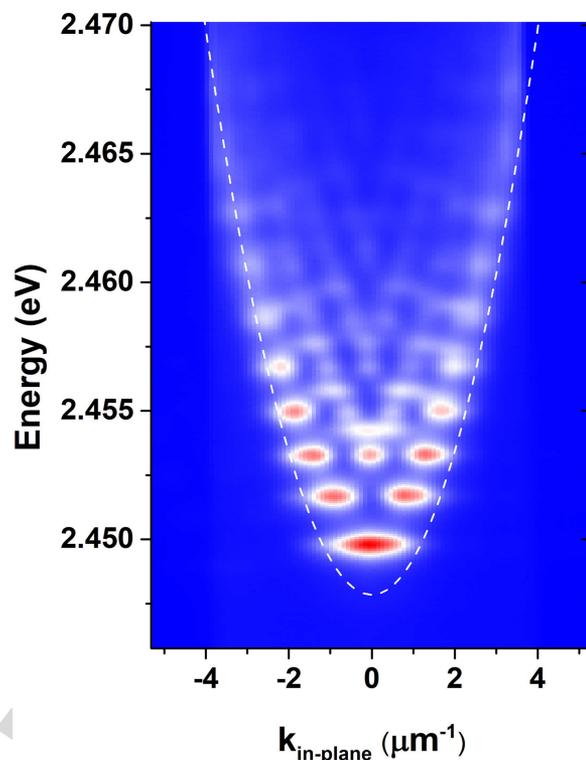


Fig. 7. k space distribution of the 3D confined optical states. The white dashed curve is the calculated mode dispersion in 2D MC without lateral confinement.

208 namely the “photonic dot” effect, can lead to distinctive phenom-
 209 ena, such as the inhibition or acceleration of the spontaneous
 210 emission in the so-called weak coupling regime [34], and the
 211 Bose-Einstein condensation of cavity polaritons in the strong
 212 coupling region [35]. The high cavity Q factor in this work is
 213 one sufficient condition to observe these split 3D confined modes
 214 and can guarantee more concentrated optical state distribution
 215 because of the narrow mode linewidth. At present, only the
 216 3D confined optical states were observed, but the electrically
 217 injected high Q factor structure in this work is also eminently
 218 suitable for other material systems. We believe that the structure
 219 is promising for 3D confined polariton devices working under
 220 strong coupling condition, if active layers with narrower exciton
 221 emission linewidth and larger exciton binding energy such as
 222 thin GaN or AlGaIn QWs are utilized.

223 In the following, the loss mechanism is systematically inves-
 224 tigated in the cavity with a high Q factor. The intensity of the
 225 light inside the cavity after one round-trip propagation can be
 226 given by

$$R_1 R_2 (1 - S_1) (1 - S_2) \cdot A e^{-\alpha_{inner} \cdot 2d} = A e^{-\alpha_{total} \cdot 2d} \quad (1)$$

227 where A is the intensity at the starting point, R_1 and R_2 are the
 228 reflectivity of DBRs, S_1 and S_2 are the scattering loss at the two
 229 end faces of the cavity, α_{inner} is inner absorption coefficient,
 230 and d is the cavity length. α_{total} is the effective loss coefficient
 231 obtained by converting all loss items into the whole cavity, and

232 can be expressed by

$$\alpha_{total} = \frac{-\ln R_1 R_2}{2d} + \frac{-\ln[(1 - S_1)(1 - S_2)]}{2d} + \alpha_{inner} \quad (2)$$

233 with the three items corresponding respectively to leaks and
234 absorption from the DBRs, the scattering loss caused by im-
235 perfect morphology of the end face of the cavity, and inner loss
236 originating from the MC layers and lateral optical leakage. Then
237 the expected Q-factor of the cavity can be decomposed into three
238 contributing terms

$$Q^{-1} = Q_{DBR}^{-1} + Q_{inner}^{-1} + Q_{scatter}^{-1} \quad (3)$$

239 The high Q factor 6039 of the fundamental mode realized
240 in this work could be attributed to the high reflectivity of the
241 dielectric DBRs, the atomic level flat end face of the cavity, the
242 lateral confinement of the optical modes induced by the buried
243 AlN LOC structure, as well as the low loss of GaN layer in the
244 green region. Q_{mir}^{-1} and $Q_{scatter}^{-1}$ could be obtained by

$$Q_{mir}^{-1} = \frac{1}{2\pi} \cdot \frac{-\ln R_1 R_2}{2d} \cdot \frac{\lambda}{n} \quad (4)$$

$$Q_{scatter}^{-1} = \frac{1}{2\pi} \cdot \frac{-\ln[(1 - S_1)(1 - S_2)]}{2d} \cdot \frac{\lambda}{n} \quad (5)$$

245
246 Where λ , n , d are the operating wavelength (495 nm), refractive
247 index (2.39), and cavity length (3819 nm), respectively. The re-
248 flectivity is 99.6% and 99.8% for the 10 and 12.5 pair TiO₂/SiO₂
249 DBRs. The scattering loss S1 (5.468e-5) and S2 (3.781e-4) are
250 calculated by $S_{scatter} = D\{1 - \exp[-(\frac{4\pi\delta}{\lambda})^2]\}$ where δ is the
251 RMS of the cavity end faces (0.29 and 0.76 nm), and D is
252 a calibration coefficient defined as the ratio between the total
253 reflectance of the top DBR versus air and that versus p-GaN [36].
254 A was calculated to be 1.02 here. Q_{mir}^{-1} and $Q_{scatter}^{-1}$ turned out
255 to be $2.122 \cdot 10^{-5}$ and $1.531 \cdot 10^{-6}$ respectively, i.e., $Q_{DBR} \sim$
256 47094 , $Q_{scatter} \sim 6.532 \cdot 10^5$. These values are much larger
257 than the measured Q factor of 6039, meaning that the loss from
258 the DBRs and surface scattering only occupy a very small part of
259 the total loss in the cavity in our case, due to the high reflectivity
260 of the DBRs and the atomic level flat cavity end face. Q_{inner}^{-1}
261 was calculated to be $\sim 1.427 \cdot 10^{-4}$ i.e., $Q_{inner} \sim 7009$, in the same
262 order of the measured Q value. So, the main loss of the cavity
263 come from the inner loss originating from the absorption of
264 cavity layers, and lateral optical leakage. The inner absorption
265 coefficient α_{inner} was calculated to be 46 cm^{-1} from the equation
266 $Q_{inner}^{-1} = \frac{1}{2\pi} \cdot \alpha_{inner} \cdot \frac{\lambda}{n}$, and mainly consists of three
267 parts including the absorption of GaN layers in the cavity, the
268 absorption due to the ITO layer and the lateral optical loss. The
269 absorption coefficient of doped GaN layer in the green region
270 is $\sim 10 \text{ cm}^{-1}$ [37], and the equivalent absorption coefficient of
271 ITO layer was calculated to be $\sim 14 \text{ cm}^{-1}$ by $\Gamma_r \cdot \alpha_{ITO} \cdot \frac{l}{d}$,
272 where Γ_r (~ 1.0 here calculated by transfer matrix method) is
273 the relative gain enhancement factor which is used to express the
274 overlap degree of the ITO layer and standing wave field in the
275 cavity, α_{ITO} is the absorption coefficient of ITO (2000 cm^{-1})
276 [37], and l is the thickness of ITO layer ($\sim 30 \text{ nm}$) used in this
277 work. Then we obtained that the absorption coefficient caused

278 by the lateral optical loss is $\sim 22 \text{ cm}^{-1}$. This is a relatively
279 small value in GaN-based vertical-cavity structures owing to
280 the lateral optical confinement by the buried AlN LOC structure.
281 Masaru Kuramoto *et al.* reported GaN VCSEL with buried SiO₂
282 LOC structure and the lateral leakage loss decreased from 73 to
283 23 cm^{-1} when the LOC structure was applied, similar to our
284 results [38]. The use of the AlN layer instead of SiO₂ in this
285 work can improve the thermal dissipation in GaN-based MC
286 with dual dielectric DBRs. The insulation layer in GaN-based
287 MC devices with dual dielectric DBR structure is embedded
288 between p-GaN and substrate. It locates at the main pathway of
289 thermal dissipation from the active region to substrate. In our
290 previous works [39], [40], we found that the SiO₂ LOC layer
291 will degrade device thermal dissipation because of the relatively
292 low thermal conductivity of 1.5 Wm/K . In contrast, AlN has a
293 much higher thermal conductivity of $\sim 200 \text{ Wm/K}$. Therefore,
294 using AlN will enable us to improve thermal dissipation inside
295 the device. But it needs to be noted that lasing action was not
296 realized in this work although a 3D confined low loss cavity with
297 a high Q factor was fabricated. At present, green VCSEL is still
298 difficult based on c-plane InGaN QWs grown on the sapphire
299 substrate because of the “green gap”, which is caused by the
300 large quantum confined stark effect and low emission efficiency
301 in the green region [30], [42]. The material gain of c-plane green
302 InGaN QWs still needs further optimization to realize lasing
303 action.

IV. CONCLUSIONS

304
305 In summary, we have demonstrated an electrically injected
306 green GaN-based MC light emitter with a high Q factor of 6039
307 which is the highest value in electrically injected GaN-based
308 MC light emitters to the best of our knowledge. The MC light
309 emitter is featured with dual dielectric DBRs and an AlN buried
310 LOC structure. The emission characteristics as a function of
311 cavity length, as well as the loss mechanism in the cavity, were
312 systematically analyzed. The high Q factor was contributed to
313 the high reflectivity of DBRs, low scattering loss of the cavity
314 end face, as well as the small lateral optical leakage induced
315 by the LOC structure. The 3D confined optical states inside the
316 cavity were also observed and studied through angle-resolved
317 measurement. This electrically injected 3D confined GaN-based
318 MC light emitter with a high Q factor is promising for the study
319 of CQED at room temperature including Purcell effect and Bose-
320 Einstein condensation of cavity polaritons.

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